

# ANALYTICAL VALIDATION OF THE YOUNG-DUPRÉ LAW FOR EPITAXIALLY-STRAINED THIN FILMS

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University of Vienna

Joint work with E. Davoli, Vienna

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Calculus of Variations*  
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FWF

Der Wissenschaftsfonds.



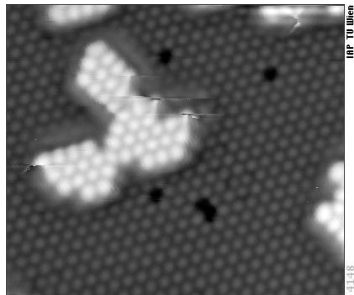
VIENNA SCIENCE  
AND TECHNOLOGY FUND

The focus is on:

- deriving reliable **variational models** for thin films deposited on substrates;
- studying the **morphology** and the **geometry** of film profiles.

Plan of the talk:

1. Description of the **deposition process**;
2. Variational formulation of the **model**;
3. **Existence** of minimizers;
4. **Properties** of minimizers.



STM image of Pt on Pt(111) by PLD, Diebold's lab, TU Wien.

# 1. THE DEPOSITION PROCESS

Thin films are grown by **Molecular Beam Epitaxy (MBE)** (or vapor deposition), and **Pulsed Laser Deposition (PLD)**.

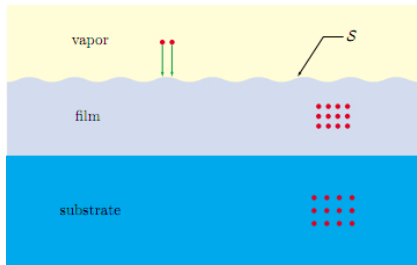
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# 1. THE DEPOSITION PROCESS

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We consider:

- **heteroepitaxy** (different elasticity properties of the 2 materials);
- the presence of a **mismatch** between the crystalline lattices;
- the **3 interfaces**: film/vapor (free), substrate/vapor and substrate/film;
- both **wetting and dewetting regimes**.



Epitaxy [Fried-Gurtin, 2004].

Different **modes of film growth**:

(VW) Volmer–Weber;

(FM) Frank–van der Merwe;

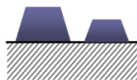
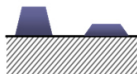
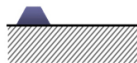
(SK) Stranski–Krastanov.

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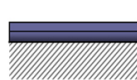
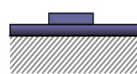
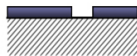
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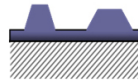
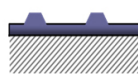
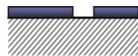
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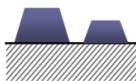
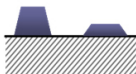
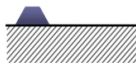
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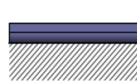
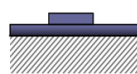
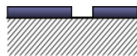
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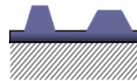
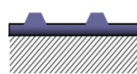
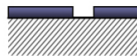
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What determines these different growth modes?

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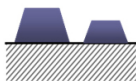
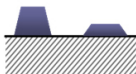
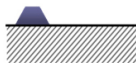
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- the lattice mismatch;
- the different surface tensions.

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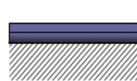
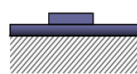
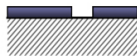
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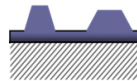
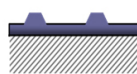
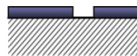
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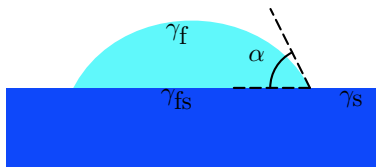
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How large is the island angle formed at the substrate?

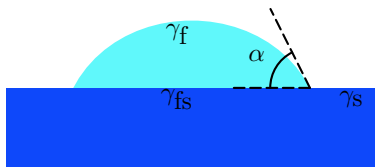




- $\alpha$  is the **contact angle** formed by island profiles with the substrates;
- $\gamma_s :=$  **substrate/vapor** surface tension;

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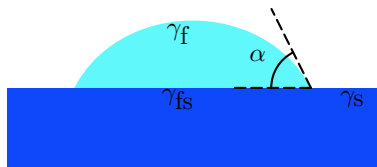
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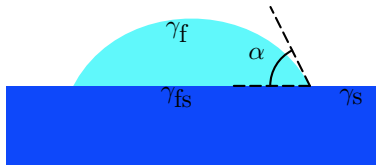
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The **Young-Dupré law** is the following:

$$\cos \alpha = \frac{\gamma_s - \gamma_{fs}}{\gamma_f},$$

with  $\alpha = 0$  if  $\gamma_s - \gamma_{fs} \geq \gamma_f$  (see [Young, 1805] and [A. Dupré-P. Dupré, 1869]).

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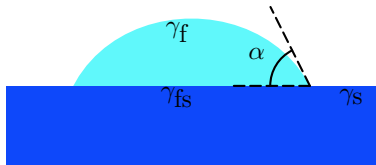
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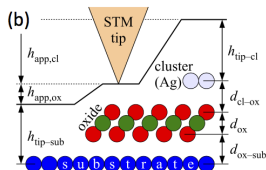
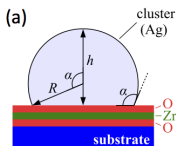
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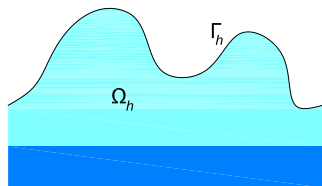
In the **experiments** contact angles are computed **from the island height and the radius** of the top facet obtained by STM.



Clusters on zirconia<sup>†</sup>. Courtesy of Diebold's lab.

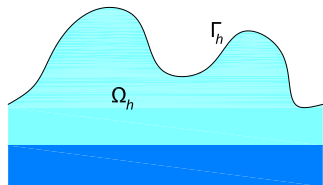
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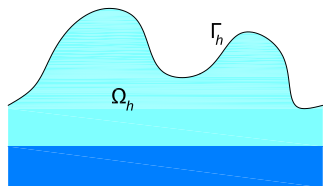
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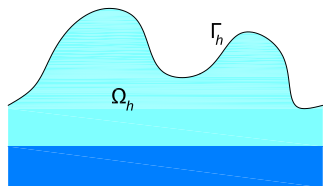
- $u : \Omega_h \rightarrow \mathbb{R}^2$  being the planar displacement;
- $\mathbf{E}u := \frac{1}{2}(\nabla u + (\nabla u)^T)$  represents the strain;
- Energy minimum occurs at the mismatch strain  $\mathbf{E}_0$  defined by

$$\mathbf{E}_0(y) := \begin{cases} e_0 \mathbf{e}_1 \otimes \mathbf{e}_1 & \text{if } y \geq 0, \\ 0 & \text{otherwise,} \end{cases}$$

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Regular configurations:

$$X_{\text{Lip}} := \{(u, h) : u \in H_{\text{loc}}^1(\Omega_h; \mathbb{R}^2), h \in W^{1, \infty}(0, \ell), |\Omega_h \cap \{y > 0\}| = V\}$$

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where

- the **elastic energy** density is defined by  $W_0(y, \mathbf{A}) := \frac{1}{2} \mathbb{C}_0(y) \mathbf{A} : \mathbf{A}$  for

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- the **interface energy** was neglected in [Spencer, 1999].

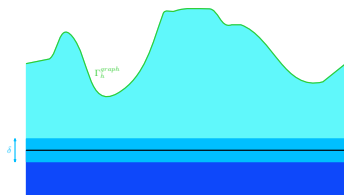
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$$\mathcal{F}_\delta(h, u) := \int_{\Omega_h} W_\delta(y, \mathbf{E}u(x, y) - \mathbf{E}_\delta(y)) \, dx \, dy \\ + \int_{\Gamma_h} \varphi_\delta(y) \, d\mathcal{H}^1$$

for a (small)  $\delta > 0$  and  $W_\delta$  defined by

$$W_\delta(y, \mathbf{A}) := \frac{1}{2} \mathbb{C}_\delta(y) \mathbf{A} : \mathbf{A},$$

where  $\mathbb{C}_\delta$ ,  $\mathbf{E}_\delta$ , and  $\varphi_\delta$  are the **regularized versions** of  $\mathbb{C}_0$ ,  $\mathbf{E}_0$ , and  $\varphi_0$  for  $y \in \mathbb{R}$ :



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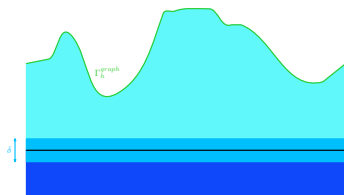
$$\mathbb{C}_\delta(y) := \frac{1}{2} \left(1 + f\left(\frac{y}{\delta}\right)\right) \mathbb{C}_f + \frac{1}{2} \left(1 - f\left(\frac{y}{\delta}\right)\right) \mathbb{C}_s + \frac{1}{2} \left(1 + f\left(\frac{y}{\delta}\right)\right) \left(1 - f\left(\frac{y}{\delta}\right)\right) (\mathbb{C}_f - \mathbb{C}_s),$$

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$$\varphi_\delta(y) := \gamma_f f\left(\frac{y}{\delta}\right) + (\gamma_s - \gamma_{fs}) \left(1 - f\left(\frac{y}{\delta}\right)\right),$$

with  $f$  some **increasing function** such that  $\int_{-\infty}^0 (1 + (f(y))^2) \, dy < +\infty$ ,

$$f(0) = 0, \quad \text{and} \quad \lim_{s \rightarrow \pm\infty} f(s) = \pm 1.$$



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$$\sup_n \mathcal{F}_\delta(h_n, u_n) < \infty \implies \sup_n \int_0^\ell \sqrt{1 + (h'_n)^2} < \infty \text{ and } \sup_n \int_{\Omega_{h_n}} |Eu_n|^2 < \infty$$

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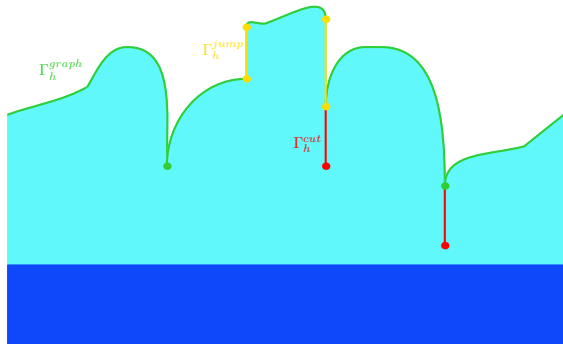
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We say that  $(h_n, u_n) \rightarrow (h, u)$  in  $X$  where

$$X := \{(u, h) : u \in H_{loc}^1(\Omega_h; \mathbb{R}^2), h \in BV(0, \ell), h \text{ is l.s.c.}, |\Omega_h \cap \{y > 0\}| = V\}.$$

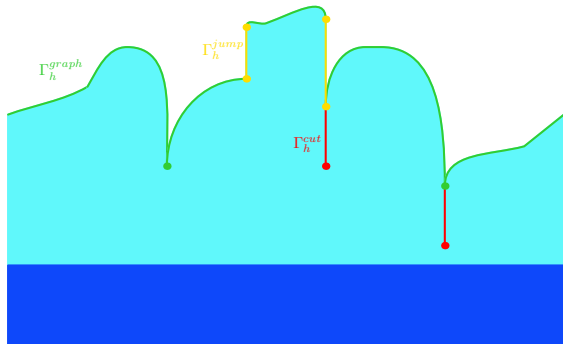
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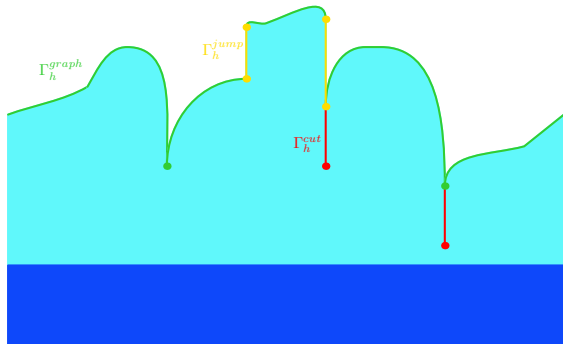


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continuous parts of  $\Gamma_h$ ,

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We now consider a sharp-interface model  $\mathcal{F}$  defined by

$$\mathcal{F}(u, h) := \int_{\Omega_h} W_0(y, \mathbf{E}u(x, y) - \mathbf{E}_0(y)) dx dy + \int_{\tilde{\Gamma}_h} \varphi(y) d\mathcal{H}^1 + 2\gamma_f \mathcal{H}^1(\Gamma_h^{cut})$$

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Notice that:

- $\mathcal{F}$  was introduced for the case  $\mathbb{C}_f = \mathbb{C}_s$  and  $\gamma_{fs} = 0$  in [Bonnetier-Chambolle, 2002] and [Fonseca-Fusco-Leoni-Morini, 2007];
- Cuts are counted twice as they are approximated by shrinking valleys;
- If  $\gamma_f \leq \gamma_s - \gamma_{fs}$ , then  $\varphi \equiv \gamma_f$ .

The energy  $\mathcal{F}$  satisfies the following assertions:

1.  $\mathcal{F}_\delta \xrightarrow{\Gamma} \mathcal{F}$  in  $X$  as  $\delta \rightarrow 0^+$ ;
2.  $\mathcal{F}$  is the relaxation of  $\mathcal{F}_0$  in  $X$ , i.e.,

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Remark on the proof:

- For the  $\Gamma$ -convergence we extend the argument in [Fonseca-Fusco-Leoni-Morini, 2007] based on the [integral formula for the relaxation  \$\bar{\mathcal{F}}\_\delta\$](#)  of the  $\mathcal{F}_\delta$  in  $X$ , i.e.,

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- Extra care is needed for the relaxation of the sharp-interface model for  $\gamma_s - \gamma_{fs} < \gamma_f$  in the construction of a [recovery sequence that matches the volume constraint](#).

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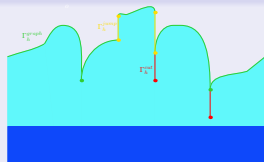
If  $(h, u) \in X$  is a minimum configuration for  $\mathcal{F}$ , then

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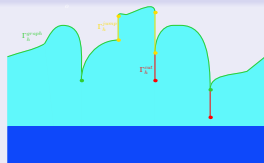
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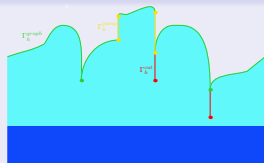
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2. **Internal-Ball condition** ([Chambolle, Larsen; 2003]): There exists  $\rho > 0$  such that for every  $z \in \bar{\Gamma}_h$  a ball  $B_\rho$  with radius  $\rho$  can be chosen so that

$$B_\rho \subset \Omega_h \quad \text{and} \quad \partial B_\rho \cap \bar{\Gamma}_h = \{z\}$$

(established by a comparison argument and the isoperimetric inequality).

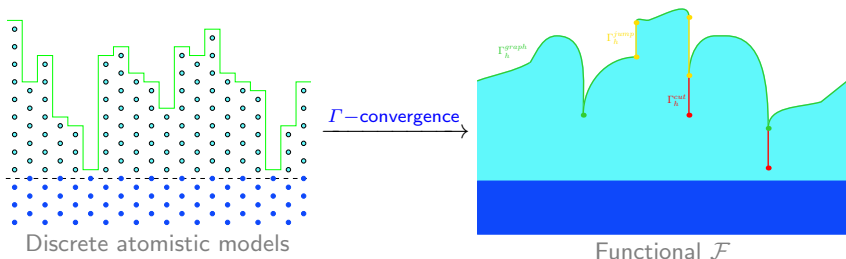
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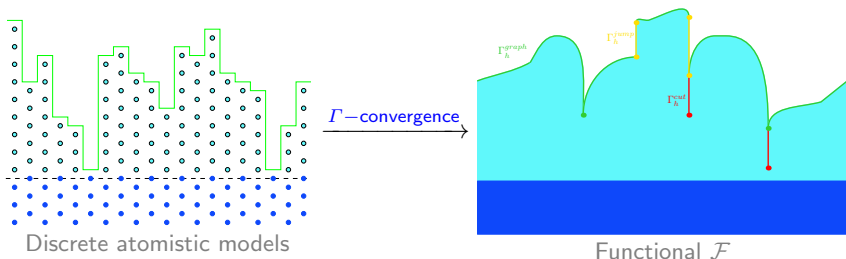
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What about the contact angles of minimal profiles of  $\mathcal{F}$ ?

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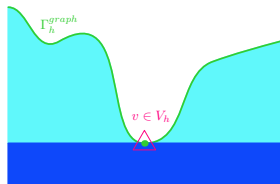
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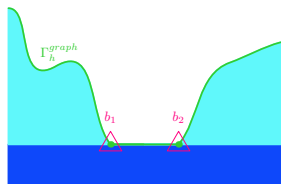
➡ Note that:

- It is a classical condition in **transmission problems** for elliptic systems;
- From [Knees, 2002]: “it seems that it [...] describes a class of composites which can sustain higher loads before breaking”;
- It implies that the **shear** and **P-wave moduli** are higher in the substrate, i.e., the **substrate is stiffer than the film**.

Two types of  $h$ -zeros in  $Z_h$  have nontrivial contact angles:



$v$  is a **valley**.



$b_1$  and  $b_2$  are **island borders**.

The following assertions hold for every  $\mu$ -local minimizer  $(u, h) \in X$  of  $\mathcal{F}$ :

1. Any nontrivial contact angle  $\alpha(z)$  at valleys and island borders  $z$  in  $Z_h \setminus (\Gamma_h^{cusp} \cup \Gamma_h^{cut})$  satisfies

$$\alpha(z) = \arccos(\sigma) \quad (\text{YD})$$

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- (YD) reduces to the **zero-angle condition** of [Fonseca-Fusco-Leoni-Morini, 2007] for the case  $\mathbb{C}_f = \mathbb{C}_s$ ,  $\gamma_{fs} = 0$  and  $\gamma_f \leq \gamma_s$ ;

## CONTACT-ANGLE CONDITIONS [DAVOLI-P., 2017]

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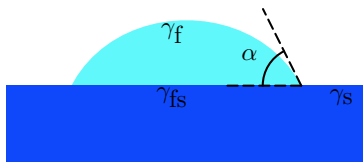
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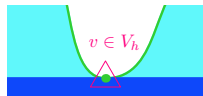
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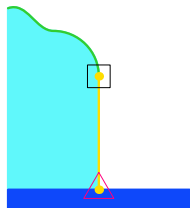
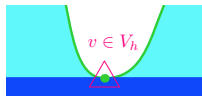
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- (YD) is the **Young-Dupré law**;



- Valleys have always zero contact angles;

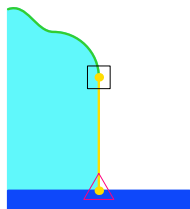
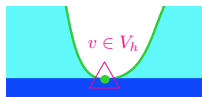


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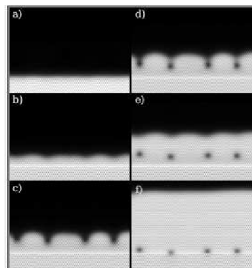
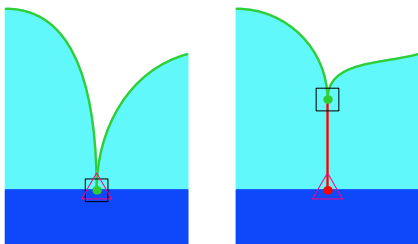




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- **Cusps** (left) and **cuts** (right) may represent dislocations at the film/substrate interface observed by experiments.



Courtesy of [Elder et al., 2007].

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There exist  $C > 0$ ,  $r_0 > 0$ , and  $1/2 < \beta < 1$  such that

$$\int_{B(z_0, r) \cup \Omega_h} |\nabla u|^2 dz \leq Cr^{2\beta} \quad \text{for all } 0 < r < r_0.$$

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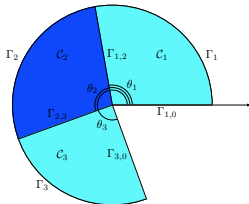
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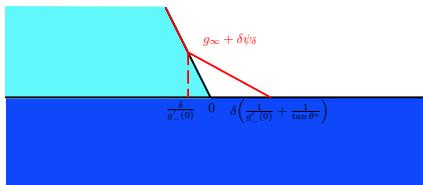
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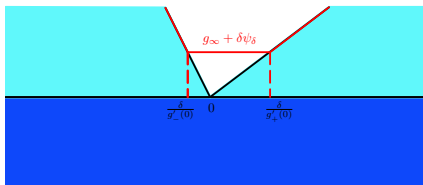
Decay obtained by contradiction and a blow-up argument in order to reduce to a transmission-problem on cones (see [Nicaise-Sändig, 1999]).



- By a suitable choice of  $\psi_\mu$  depending on the point  $z_0$  we compare with the optimal angle (red profile):



Island borders, red profile with YD angle.



Valleys, flat red profile.

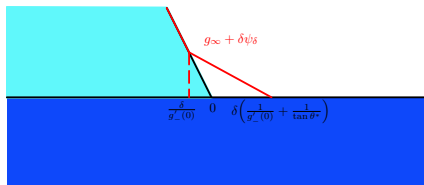
Therefore,  $S_n := S_n(h + \mu\psi_n) - S_n(h) \rightarrow S(\arccos \sigma, \alpha)$  and

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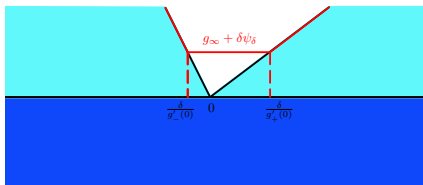
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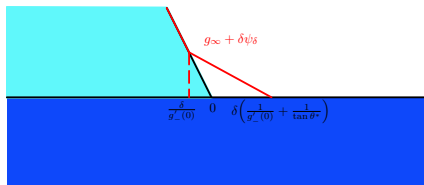
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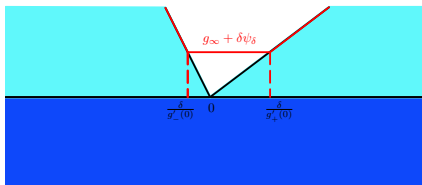
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- (i) Equilibrium contact angles are not impacted by elastic field and depend **only on surface tensions**;

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## REMARK

- Equilibrium contact angles are not impacted by elastic field and depend **only on surface tensions**;
- If  $\gamma_f \leq \gamma_s - \gamma_{fs}$ , then there is a **wetting layer** (FM and SK modes are preferable to VW);
- VW occurs** if and only if  $\gamma_f > \gamma_s - \gamma_{fs}$ .

## FURTHER REGULARITY [DAVOLI-P., 2017]

Every  $\mu$ -local minimizer  $(u, h) \in X$  of  $\mathcal{F}$  is such that

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Finally, we also have that the Euler-Lagrange equation

$$k_{\varphi, A_h} = \tau_{A_h} (W_0(\cdot, \mathbf{E}u(\cdot) - \mathbf{E}u_0)) + \lambda_0 \quad \text{on } A_h,$$

holds for  $\mu$ -local minimizer  $(u, h) \in X$ , where:

- $k_{\varphi, A_h}$  is the anisotropic curvature of  $A_h$ ;
- $\tau_{A_h}(\cdot)$  is the trace operator on  $A_h$ ;
- $\lambda_0$  is a suitable Lagrange multiplier.

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THANK YOU FOR YOUR ATTENTION!